



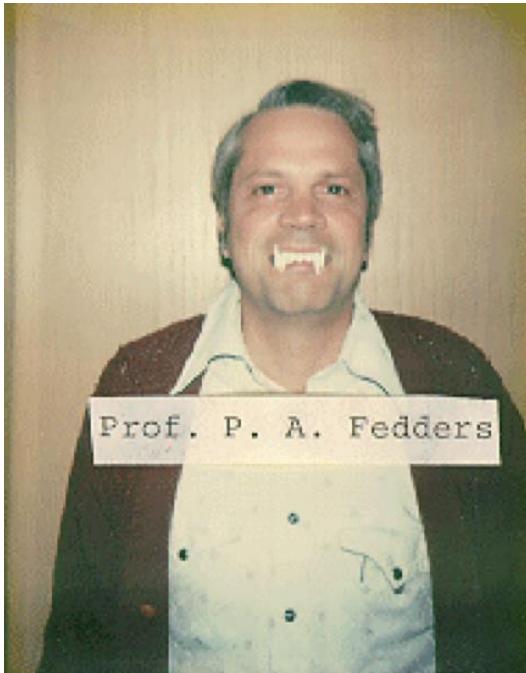
# Simulations of the Staebler-Wronski effect

*attempts at an atomistic picture*

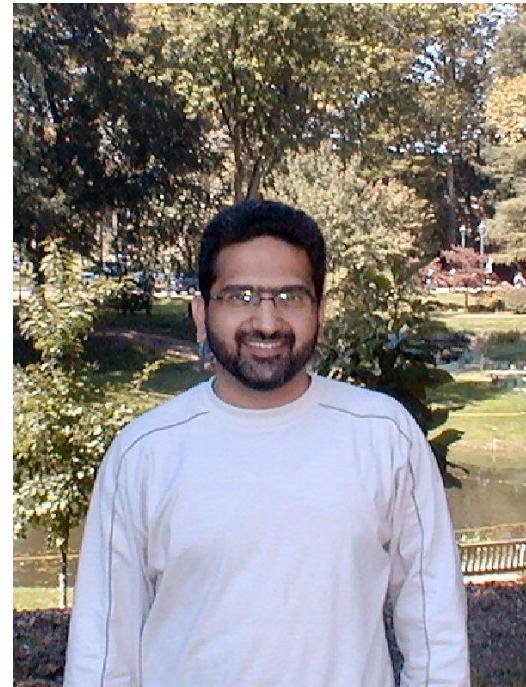
David Drabold, F. Inam

Ohio University & Trinity College,  
Cambridge

# Collaborators



Distinctive personality who started these approaches, *ca.* 1992.



F. Inam – PhD student

# Roadmap

- Background material on topology of a-Si:H, electron states, band tails and the electron-phonon coupling.
- Our approach
- Where we are, what to do.

# A vast field

- We know a lot about a-Si:H from years of experiments and theory. Almost too much (10,434 hits on WOS as of March 13 at 11.49AM)!
- The work of synthesis is therefore especially important (eg, *thank you* H. Fritzsche, B. Street, M. Stutzmann, W. Fuhs...)

# Some background from simulation

- Nuts and bolts of simulations
- Structure of a-Si:H
- Electrons in a-Si:H; **filamentary states**
- Electron-lattice interaction
- The static and dynamic role of H in the network

# Simulation nuts and bolts

- Cells with 64-1000 atoms
- Periodic Boundary Conditions
- Code: SIESTA [DFT, local basis set, double-zeta polarized (13 orbitals per site), PBE GGA]
- Tests with VASP [DFT, plane wave basis set]

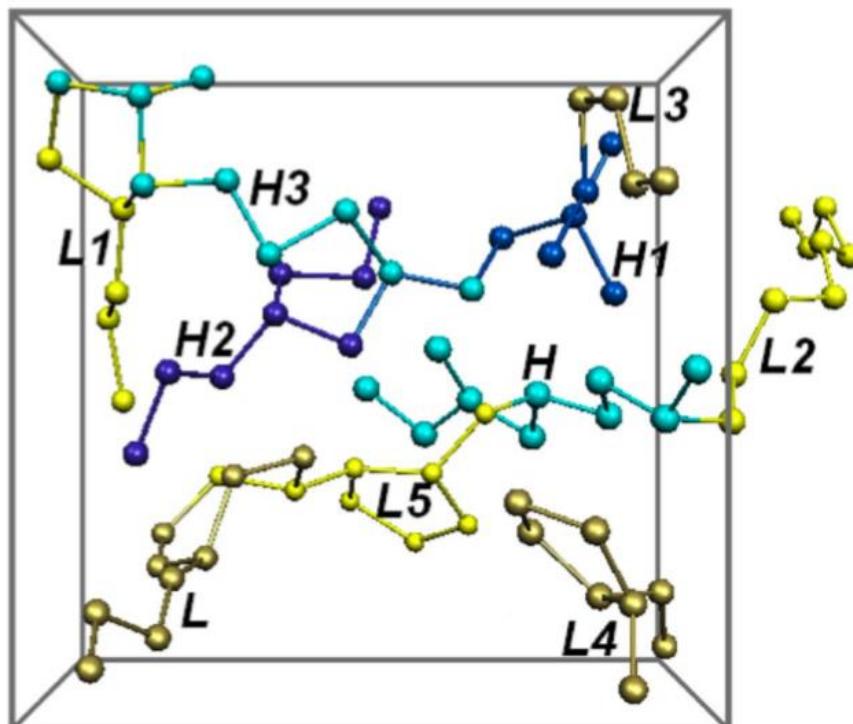
# Hidden features in a-Si structure

String theory is for us too...

- Tail eigenstates in the most realistic models are on 1-D filaments of long and short bonds we name electron filaments (long bonds, conduction; short bonds, valence).
- There are structural filaments of long and short bonds present in the models.
- The Urbach (exponential) tails are associated with these filaments.

# Two filament systems

- Electron filaments



PRL 100 206403 (2008)

- Structural filaments

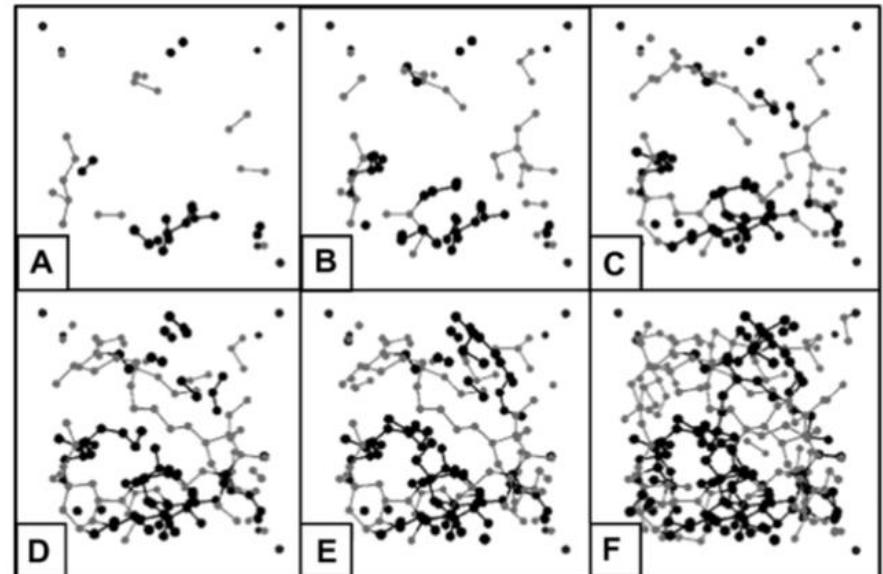
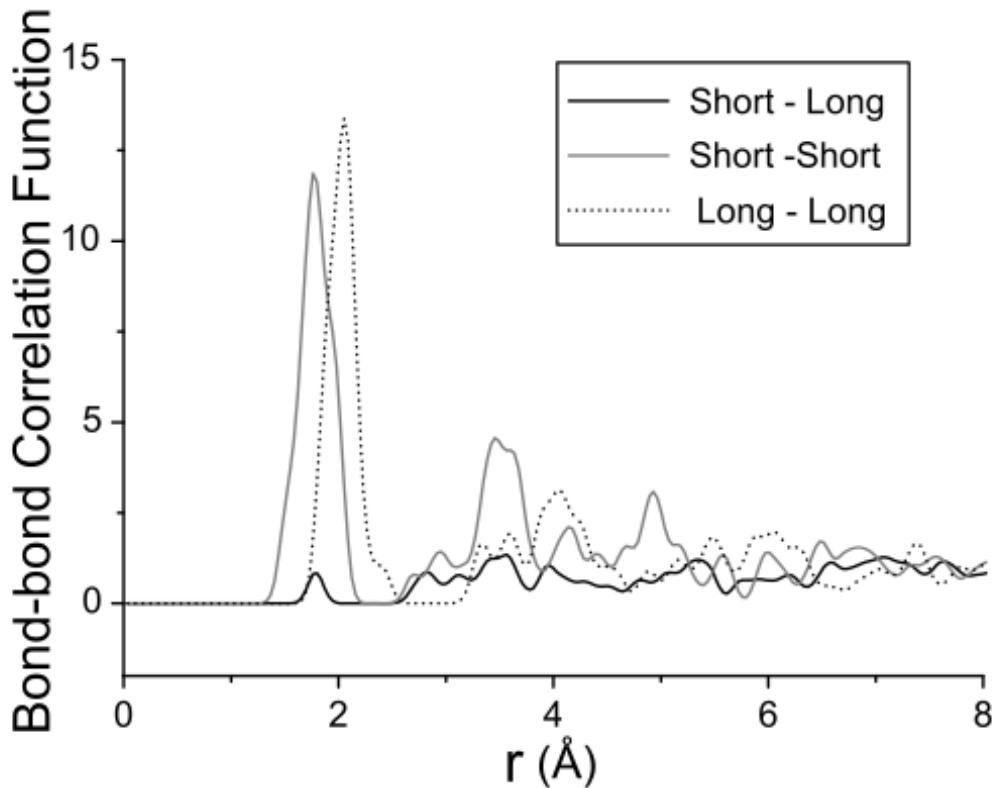


Fig. 2. (A) 1%, (B) 2%, (C) 3%, (D) 4%, (E) 5% and (F) 8% shortest(dark) and longest (light) bonds of model  $M_1$ .

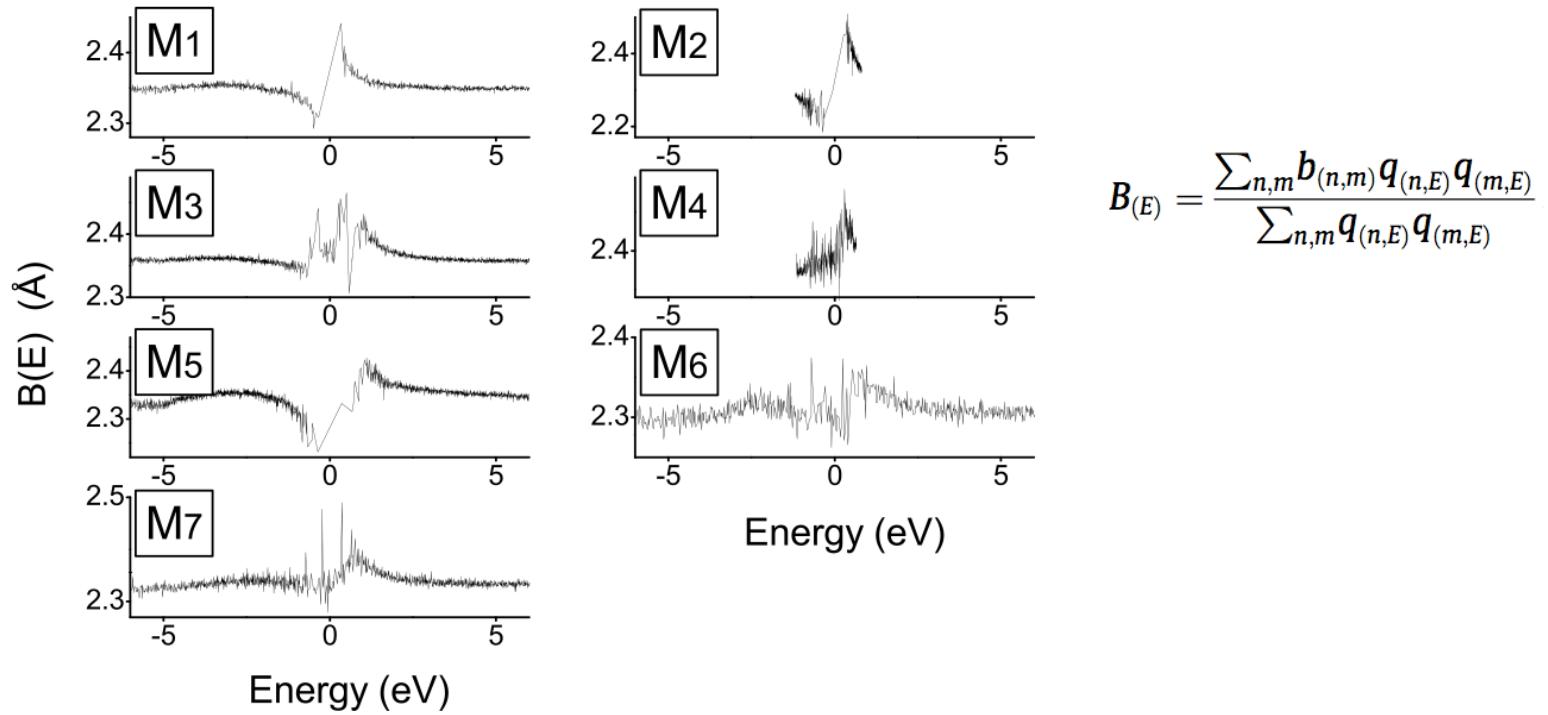
JNCS 354 3480 (2008)

# Correlation functions (l-l, s-s, s-l)



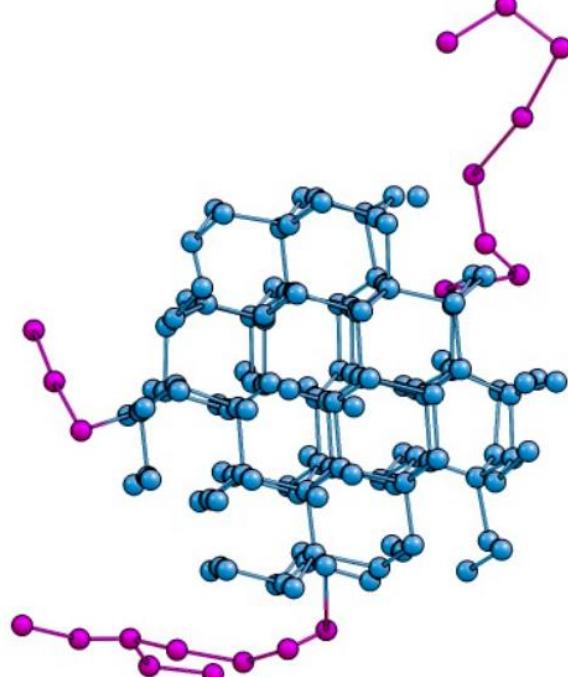
In 512-atom DTW model  
strong correlation of long bond  
to long, short to short.

# Bondlength decomposition of tail states as a function of energy

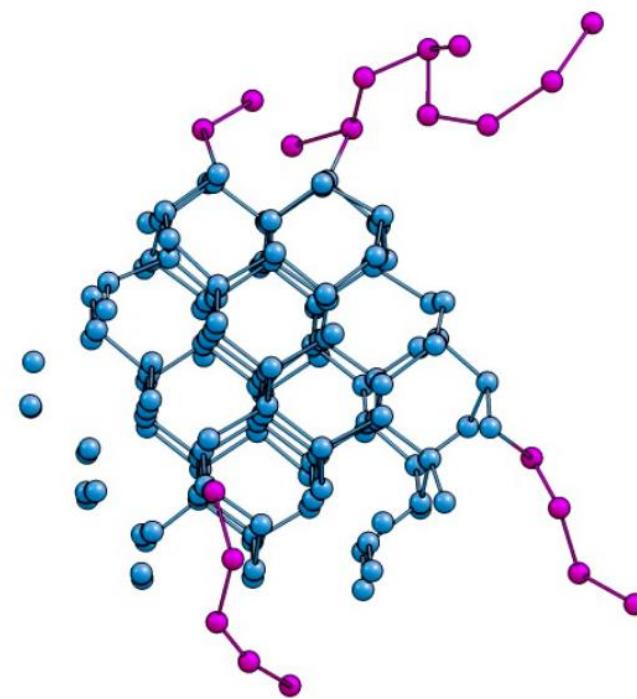


# Electron strings and crystallites: Si

*the role of inhomogeneities*



HOMO



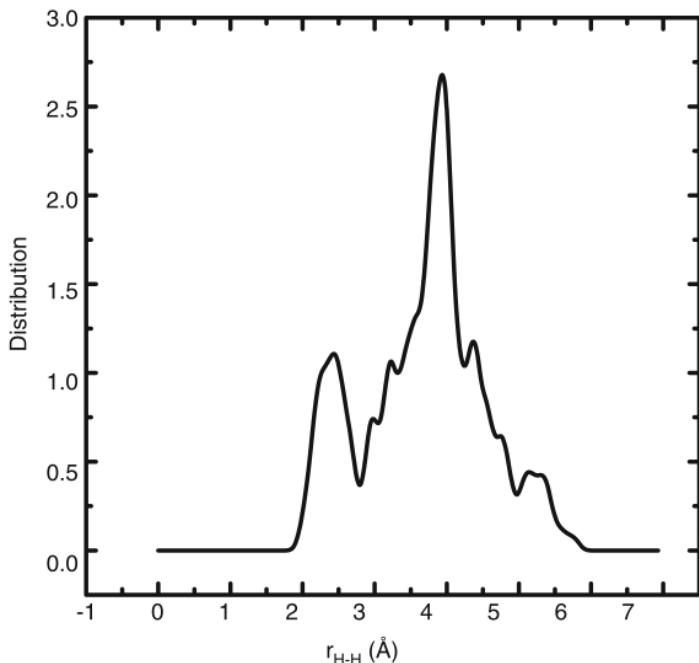
LUMO

1000 atom cell: CRN+123-atom crystallite embedded; only atoms in xtal and atoms contributing to HOMO and LUMO states are shown. (*S. Chakraborty, unpublished*)

# Structure of a-Si:H

- Wooten-Weaire-Winer CRN provides excellent model for a-Si (*sans* H); recently we have made a-Si:H without “bias”: release unbonded atomic H into network, let it “go”

J. Phys.: Condens. Matter **21** (2009) 084207



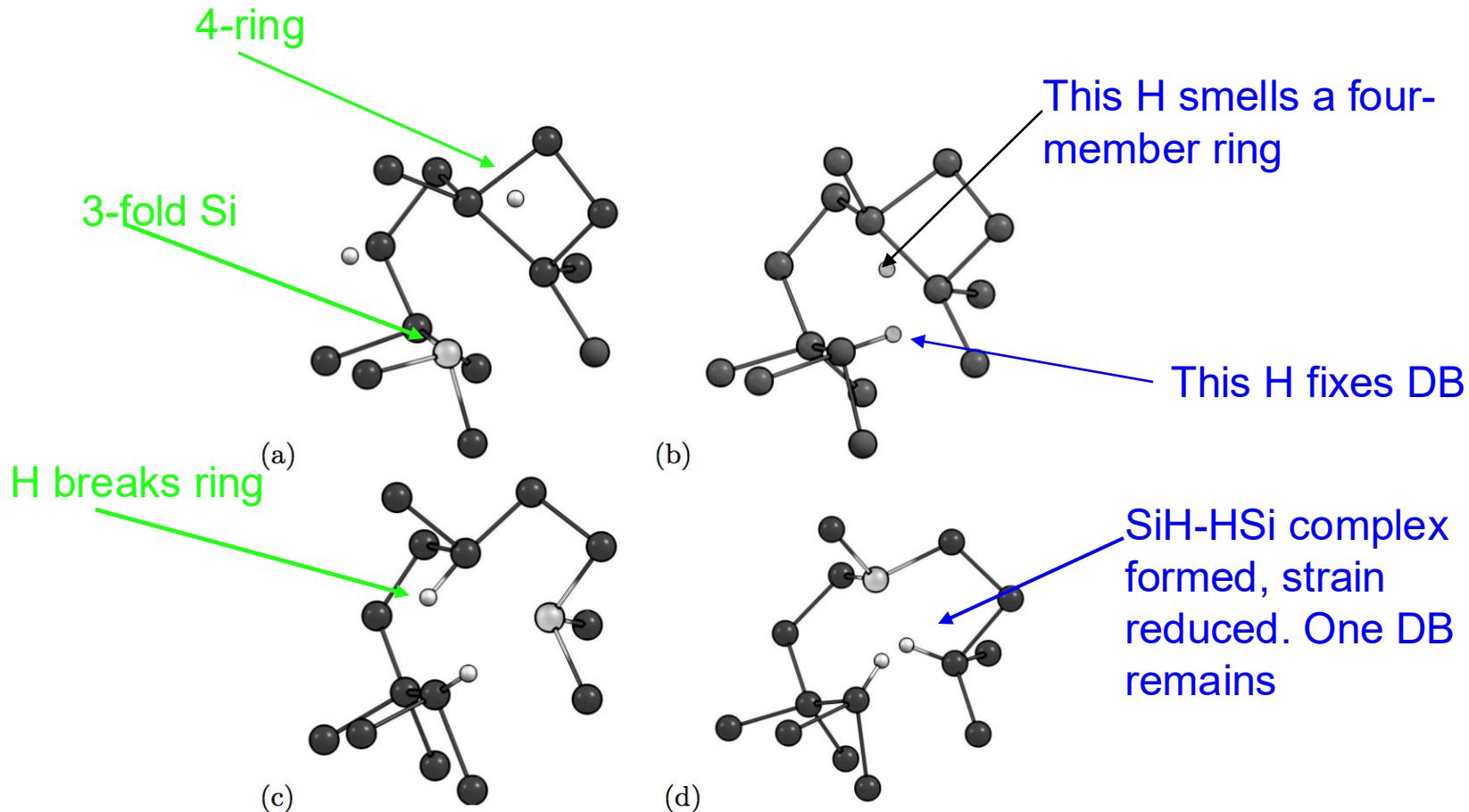
After lots of bond forming/breaking: creation of stable distribution of clustered and diffuse protons (*broad and narrow NMR lines*)!

**Figure 14.** Distribution of minimum H–H distances in amorphous Si<sub>216</sub>H<sub>24</sub> model averaged over 0.30 ps at  $T = 300$  K.

# Voids: H the strain detector/killer

S. Chakraborty, DAD PRB **79** 115214 (2009)

Start with relaxed divacancy, release the H, kill the four-member ring:



# Example of H dynamics with heterogeneous system: H<sub>2</sub> Formation in large void interior

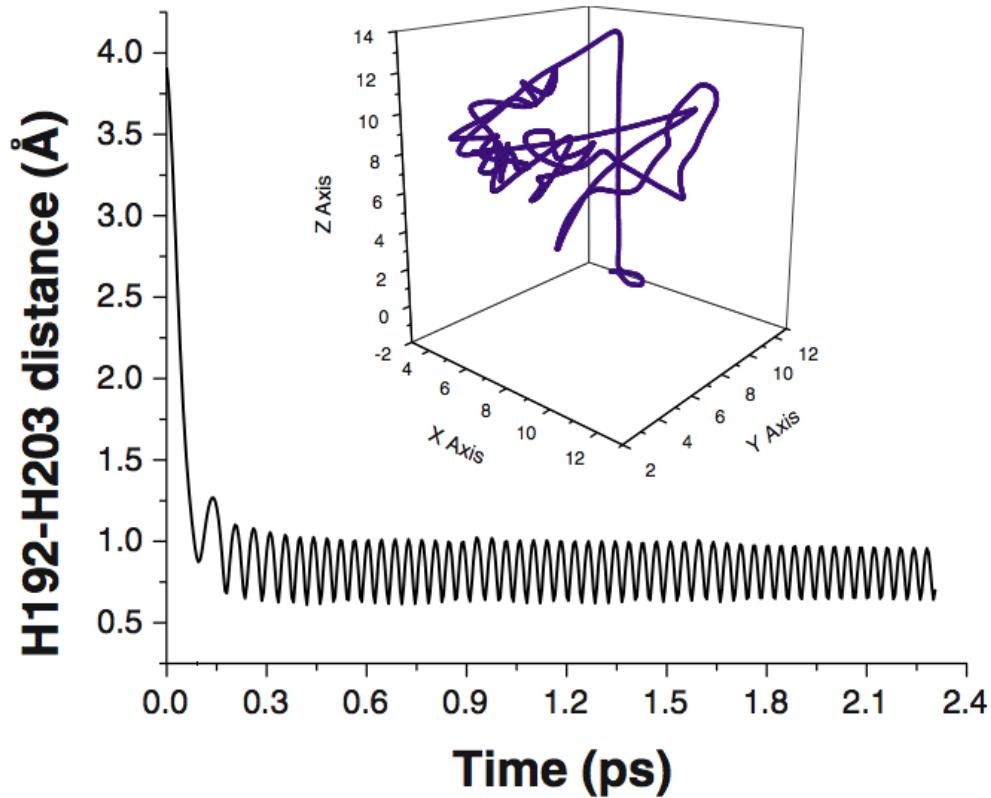
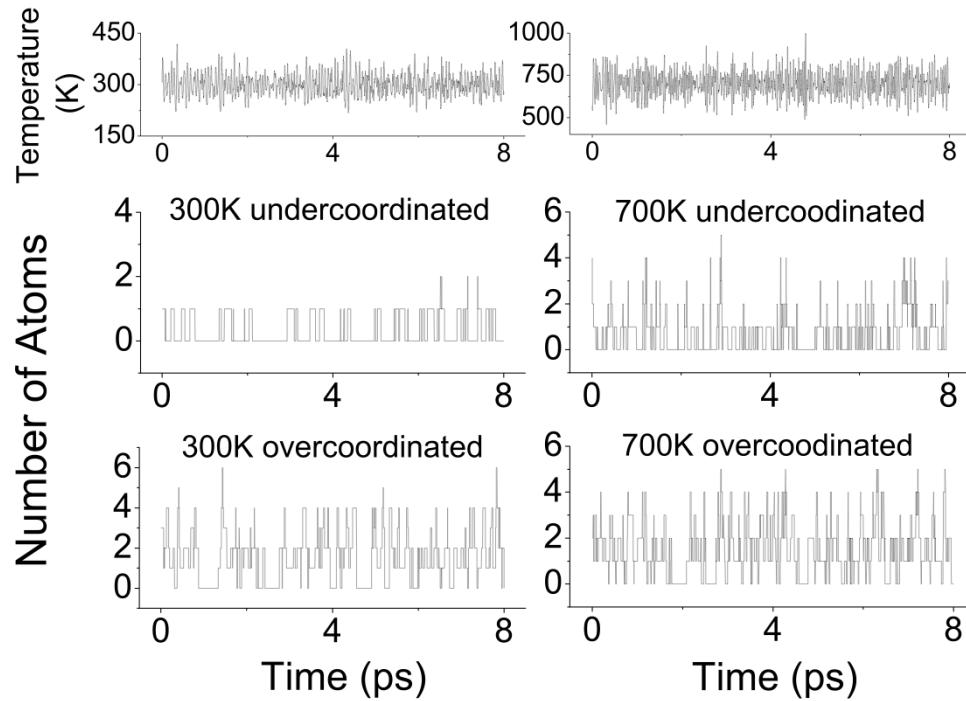


FIG. 8. (Color online) Plot of H-H distance between H<sub>192</sub> and H<sub>203</sub> showing the formation of the molecule. The trajectory of the two H in the H<sub>2</sub> molecule is shown in the inset. After H<sub>2</sub> formation, motion is confined to the void interior.

# Coordination fluctuations



Constant  $(T, V)$  simulations: SIESTA.  $R_c=2.75\text{\AA}$

J. Non-Cryst. **354** 2149 (2008)

# RMS fluctuations: Si ions

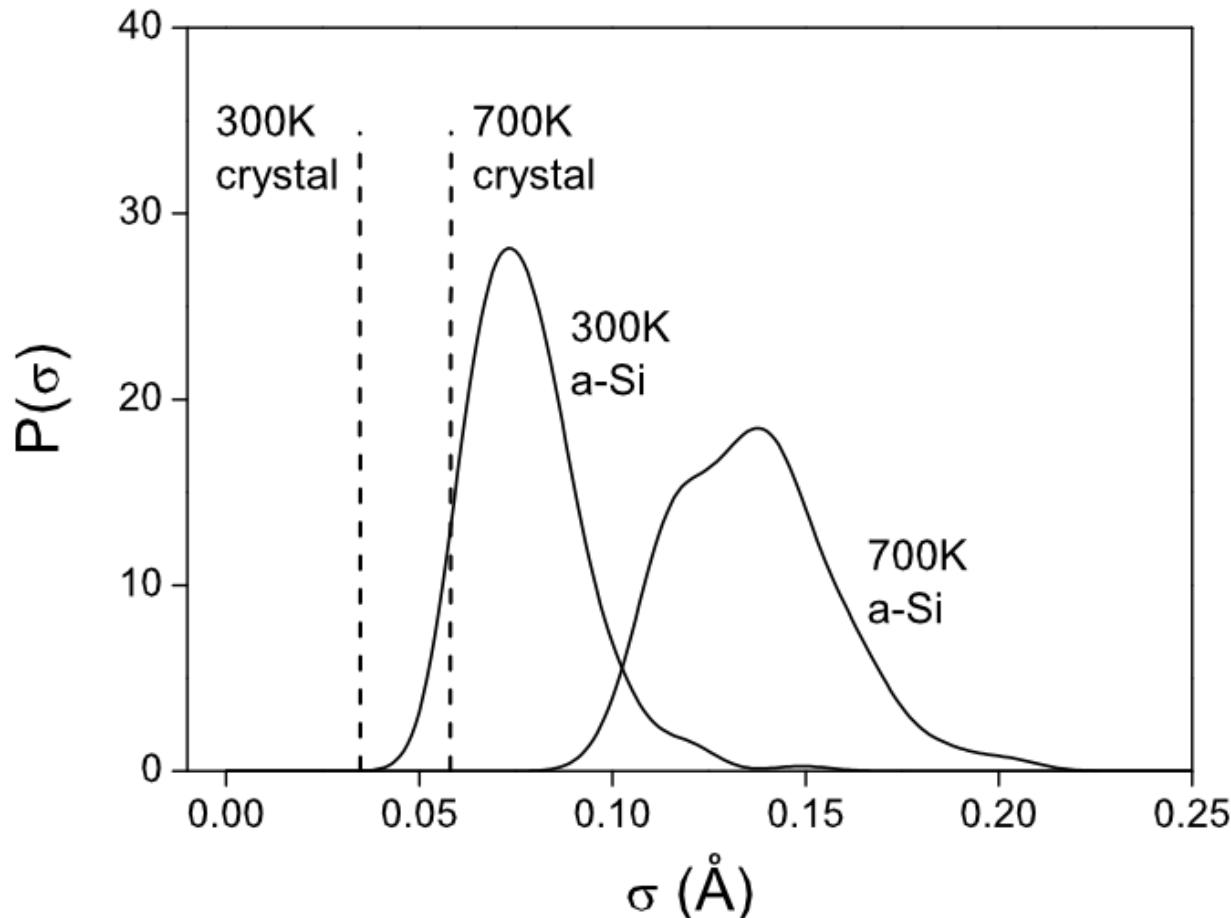
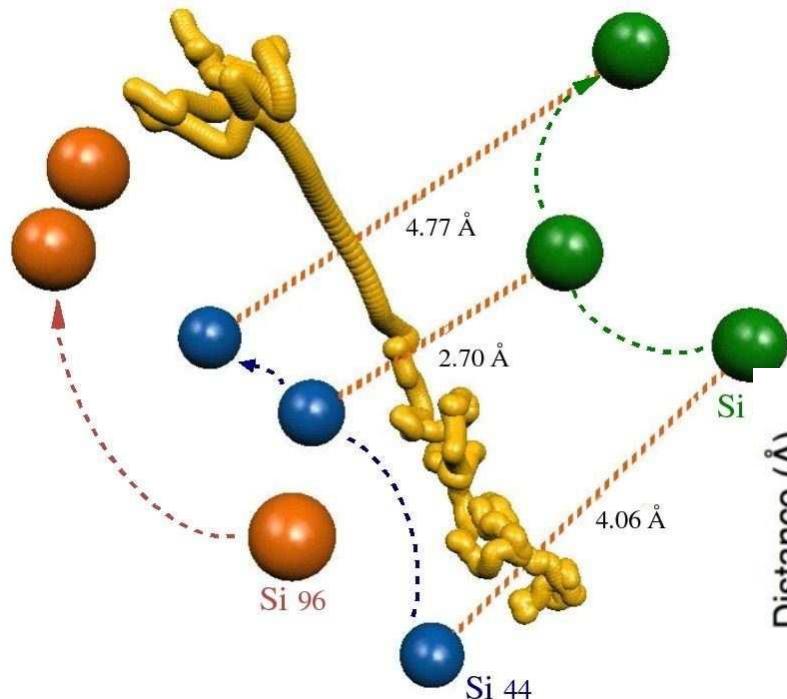


Fig. 3. Distribution of root-mean-square fluctuations  $\sigma$  in the ionic positions in 216-atom model at  $T = 300$  K and  $T = 700$  K, for both diamond and a-Si and constant  $T$  simulation, using empirical potential [25]. The data was acquired over 200 ps.

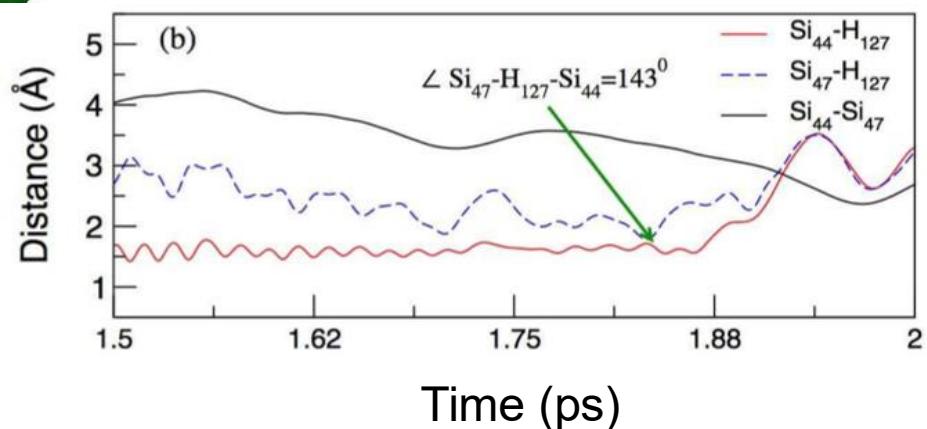
# H dynamics: Fluctuating Bond Center Detachment “FBCD”

*Converting bonded H to diffusing H*



Explicit example. Yellow worm: path of  $H_{127}$

1. H passivates DB on  $Si_{44}$
2. H becomes BC when  $Si_{47}$  “transits”
3. BC H hops, bonds to  $Si_{96}$



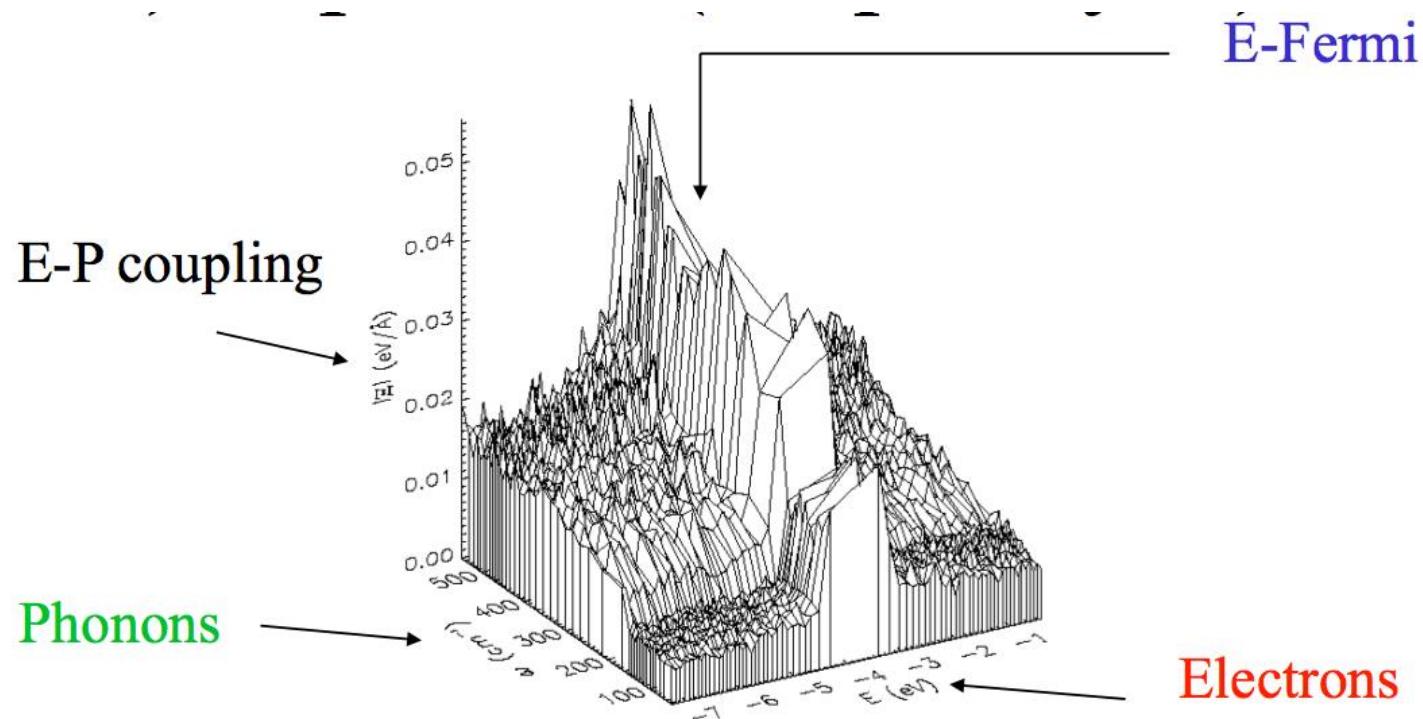
# FBCD: Discussion

- Highly temperature-dependent (at 300K, one event, 1000K, nine events in 6.25 ps)
- Reminiscent of a mechanism of Su and Pantiledes based upon “floating bonds”. FBCD is more general.
- Connected to SWE (??) -- if network is heated in any way, will stimulate H motion.

# Electron-phonon coupling and SWE

- Light affects the electronic structure of the material
- The modified electronic structure leads to changes in network structure and dynamics. Somehow, in fact, to new defects.
- Thus E-P coupling  $\Xi_n(\omega)$  (electron  $n$ , phonon  $\omega$ ) is a piece of the puzzle needed to model SWE.

# Electron-phonon coupling is large for localized states

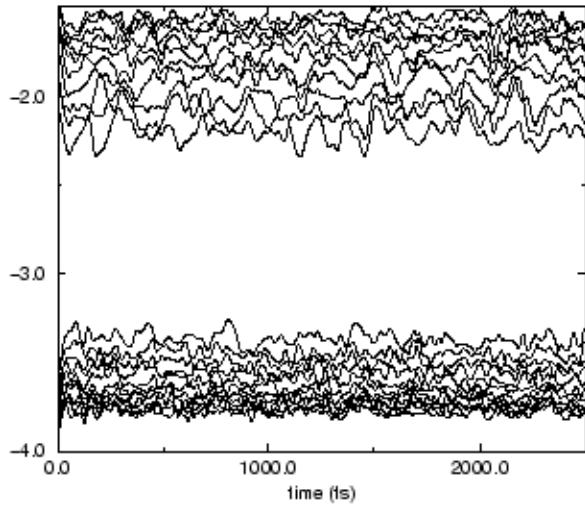


$$\Sigma_n(\omega) = \sum_{\alpha} \langle \psi_n | \partial H / \partial R_{\alpha} | \psi_n \rangle \chi_{\alpha}(\omega)$$

Couple *electron n* (energy  $E$ ) and *phonon  $\omega$*

# E-P coupling: thermally induced fluctuations in the spectrum

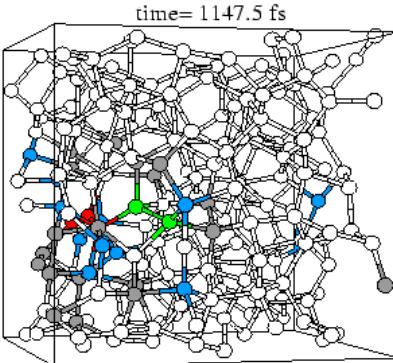
Energy eigenvalue



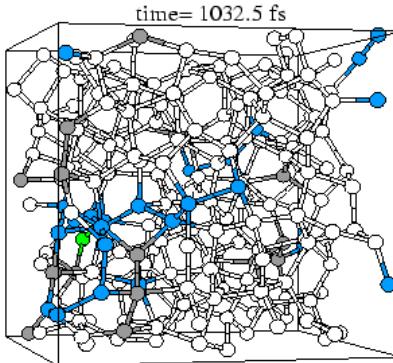
Thermally-induced fluctuations in Kohn-Sham eigenvalues (300K). 216-atom cell.

$\sigma \gg kT!$

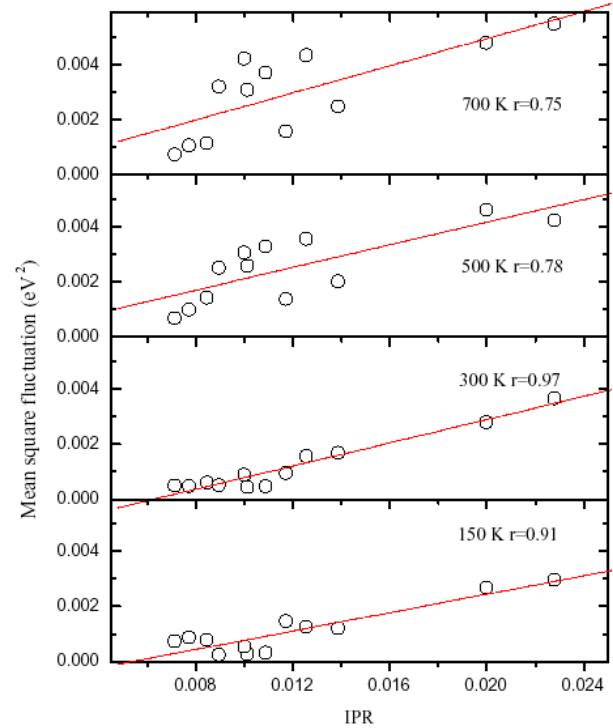
(a) A snapshot of the LUMO state:



(b) A snapshot of the LUMO state:



Fluctuations in CBE eigenvector



Correlation between IPR (localization) and RMS fluctuations.

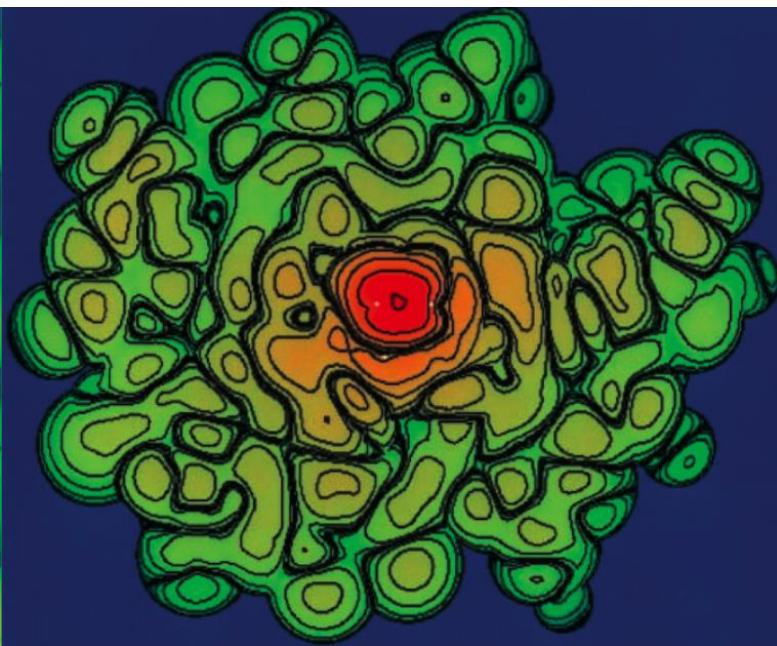
# So what?

- Only localized electron states have large coupling to lattice. *No SWE in crystal or in a material with so many defects that impurity bands form.*
- Thermally-induced variance of eigenvalue  $n$  is proportional to the localization “IPR”:  $\langle \delta\lambda_n^2 \rangle \propto IPR_n$   
Also for squared E-P coupling:  $\Xi_n^2 \propto IPR_n$
- Links a **T=0 property** (localization of a state) to **T>0 property** fluctuations in eigenvalues.

# Locality of interactions in a-Si

Kohn's *Principle of Nearsightedness* for a-Si

- Calculations of the density matrix<sup>1</sup> or generalized Wannier functions<sup>2</sup> provide quantitative information about locality of interactions in a-Si:  $\rho(\mathbf{x}, \mathbf{x}') = 2 \sum_{n-occ} \psi_n^*(\mathbf{x}) \psi_n(\mathbf{x}') = 2 \sum_n w_n^*(\mathbf{x}) w_n(\mathbf{x}')$
- Asymptotic decay is exponential:  $\rho(\mathbf{x}, \mathbf{x}') \sim \exp(-\gamma |\mathbf{x} - \mathbf{x}'|)$



Wannier function in 4096-atom WWW model of a-Si: decay is exponential,  $\gamma \sim 0.45/\text{\AA}$ . *Similar to crystal!*

*Thus,  $\sim 10\text{\AA}$  is the range of interactions in a-Si.*

<sup>1</sup>X. Zhang, DAD Phys. Rev. B **63** 233109 (2001).

<sup>2</sup>U. Stephan, R. M. Martin, DAD Phys. Rev. B **62** 6885 (2000).

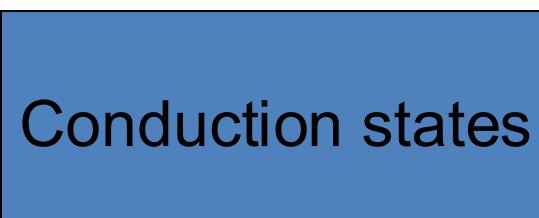
# SWE: simulation technology

- Computer models genuinely consistent with a-Si:H (structure, phonons, electrons/optics)
- Reasonable approximations for wave functions, electron-phonon couplings, charge densities, spin densities.
- Density functional methods do quite a good job on Si-H systems -- in the ground state.
- Care is needed in approximations (Chris) -- use SIESTA and DZP basis, with PBE GGA.

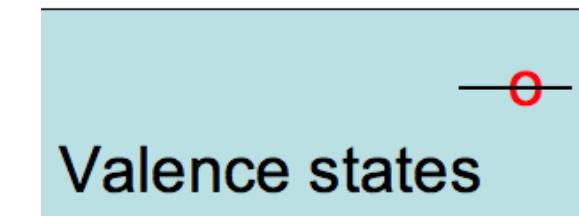
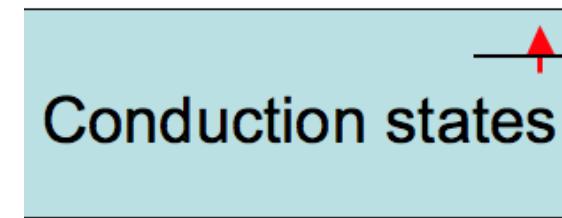
# What we can -- and cannot -- do

- We *cannot* offer credible excited states in this framework.
- We *cannot* “properly” represent the light-solid interaction (*eg*, the quantum process of light-inducing electronic transitions).
- Non-adiabatic dynamics is possible but difficult for large systems like ours.
- Can’t simulate nanosecond (or longer) dynamics directly.

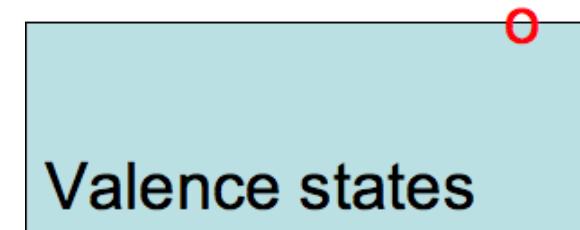
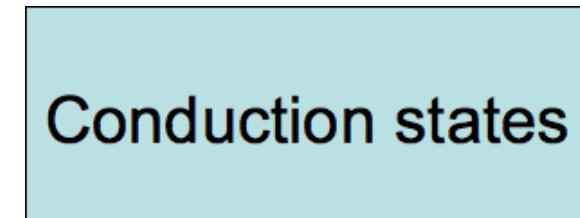
# A model for light-induced changes



$t=0$  electronic ground state



$t=t_0$ : light creates e-h pair



$t=t_0+\Delta$ ; system thermalizes, leaving uncorrelated e and h.

# Discussion

- Light creates e-h pair; this thermalizes in time  $\Delta$  with hole near VB edge (state  $\psi_h$ ), electron near CB edge (state  $\psi_e$ ).
- It is likely that  $\psi_h$  and  $\psi_e$  are localized.
- It is likely that  $|\mathbf{x}_h - \mathbf{x}_e| > 10\text{\AA}$ ; the electron and hole are uncorrelated: this is *not* an exciton (an e-h bound state).

# Discussion (continued)

*Thus, it is reasonable to model the post-thermalization dynamics by local volumes with holes and other volumes with electrons.*

*In a given ca.  $10^3$  atom cell, it is reasonable to model SWE by ground state calculations with extra or missing electrons, *not both in the same small cell.**

This model suggests that charge injection and light-induced changes are closely related.

*In our model: *hard excited state problem* = *multiple uncoupled ground state problems with different charge states.**

# Questions

- Are occupation changes in valence or conduction tail more likely to cause bond breaking/switching?
- How is H bonding and dynamics influenced by occupation changes?
- How are the results of our simulation the same (and different) from SWE, and charge injection experiments or even experience of doped systems?
- When do our approximations fail?

# First attempt: no H

PAF, Fu, DAD PRL **68** 1888 (1992)

Using simplified DF Hamiltonian on strained 63-atom cell: Change in charge state of well-localized leads to quick succession of bond breaking, highly non-local (after  $\sim 300$  fs, changes across cell)

TABLE I. Essential properties of one series of computer-simulated light-induced defects.

	Configuration				
	0	1	2	3	4
Number of geometrical defects	2	2	4	4	4
Time in fs	0	100	200	400	200
Number of spectral defects	3	3	6	5	6
Temperature (K)		200	200	200	300

# Evidence of local heating

Abtew *et al.*, JNCS 354 2149 (2008)

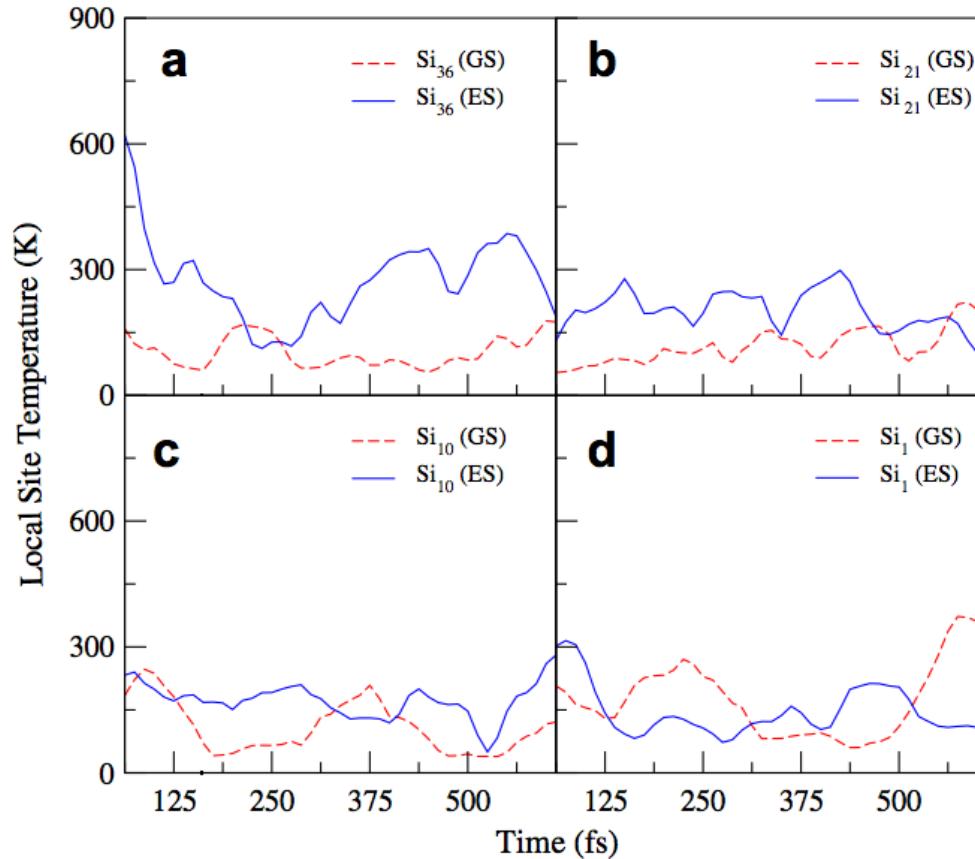
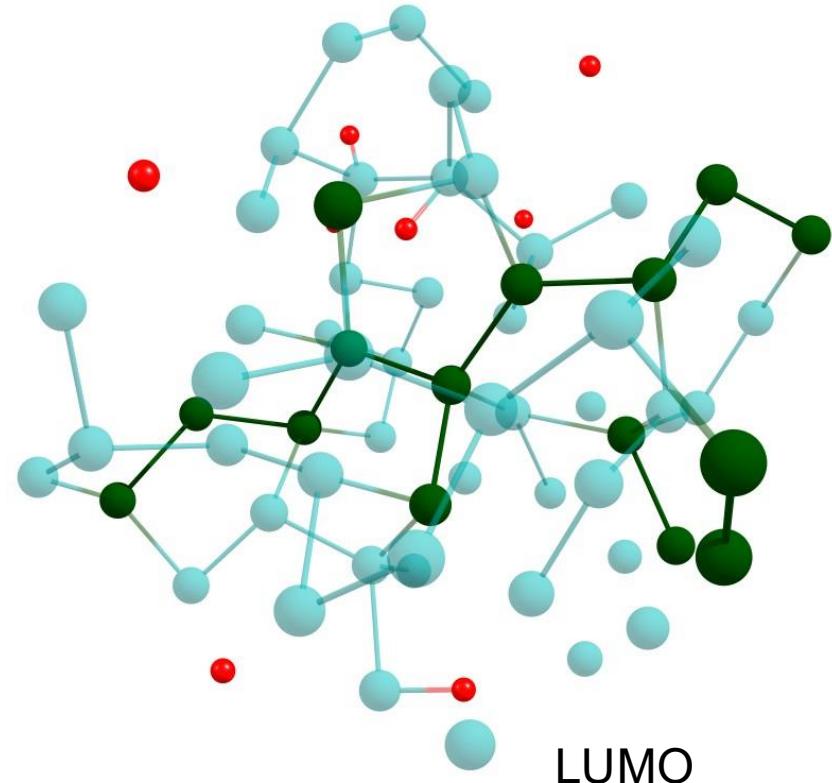
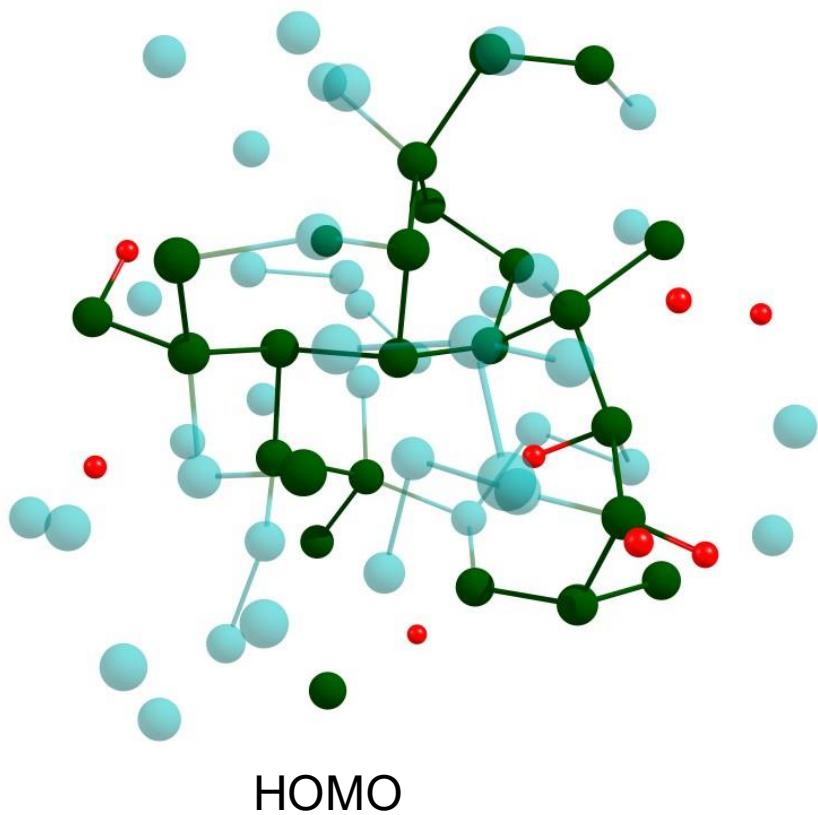


Fig. 5. Local temperature (kinetic energy) at dangling bond site ( $\text{Si}_{36}$ ) and other sites unconnected with the localized state centered on  $\text{Si}_{36}$ . GS means electronic ground state, ES, excited state.

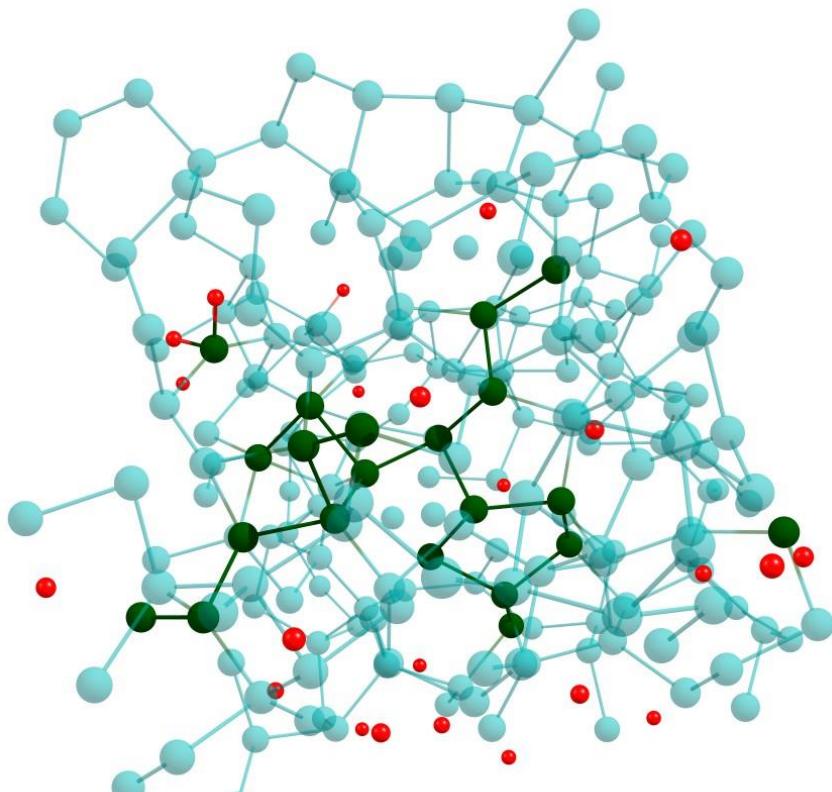
# New models

- Earlier models: H introduced manually “God’s Scissors” -- to repair defects [have recently learned of “Marshall’s (Stoneham) hand”].
- If one simply releases free H into the network, we eventually get a reasonable distribution of clustered and dilute H -- both NMR lines. Have 64, 216 atom models.
- We use these models in the rest of the talk.

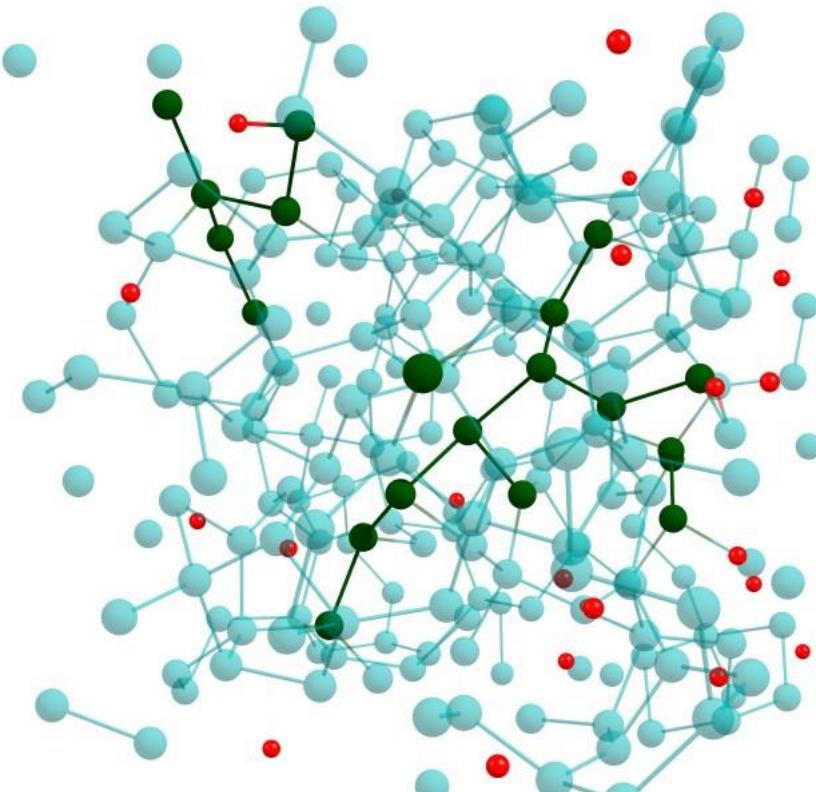
# Frontier orbitals, small cell $\text{Si}_{64}\text{H}_8$



# Frontier orbitals, large cell $\text{Si}_{216}\text{H}_{24}$



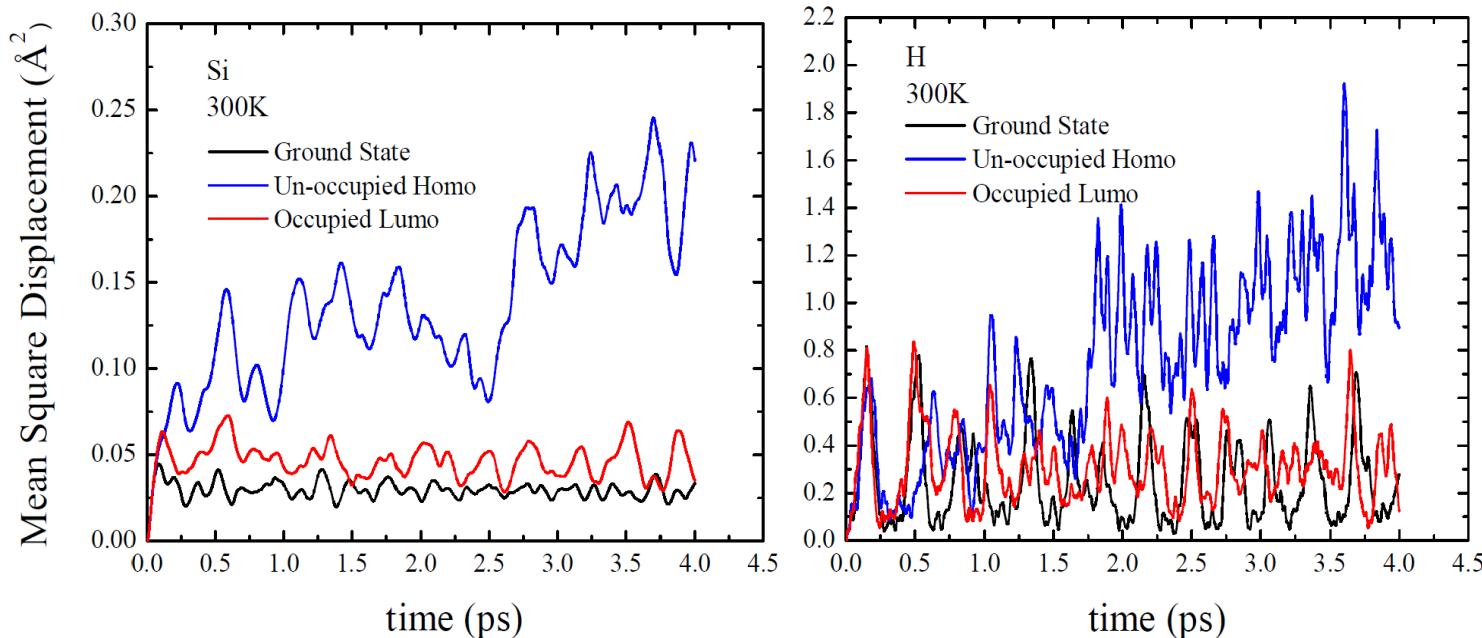
HOMO (valence)



LUMO (conduction)

# Dynamics of varied charged states

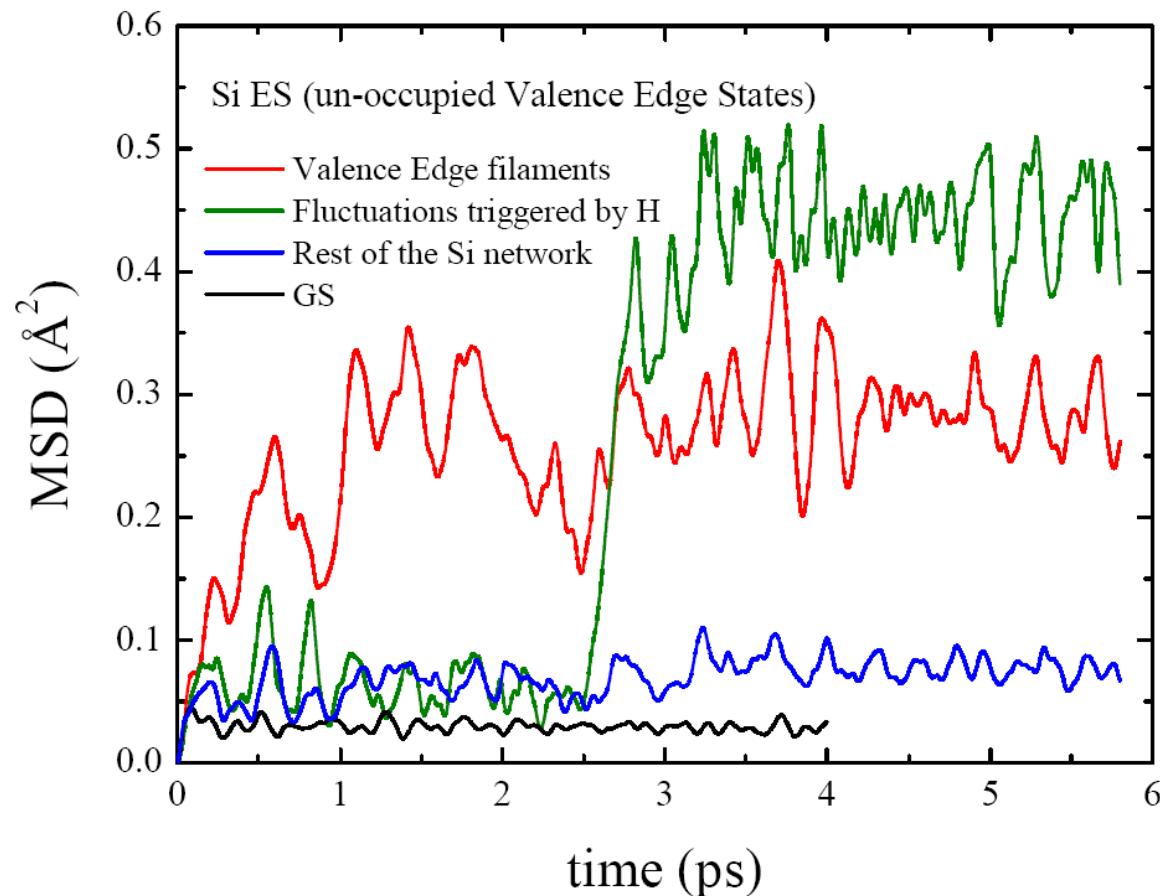
- Unoccupied valence edge states (holes) exhibit enhanced thermal fluctuation compared to occupied conduction edge states (excited electrons)<sup>1</sup>. *H dynamics is more affected by holes.*



<sup>1</sup> Seems consistent with the charge injection experiments *W. Den Bower et al., J. Non-Cryst. Solids* 66:363–68 (1984)

# Local thermal fluctuations (more local heating)

Change in the occupation of localized states (holes) gives rise to local thermal fluctuations which may spread further through H mediation.



# Conclusions

- The eigenvectors near the Fermi level are mostly associated with filaments of connected long (conduction) and short (valence) bonds.
- The electron-lattice coupling is large for localized states, and for a sufficiently strained network, occupation changes can cause bond changes.

# Conclusions

- Models with isolated and clustered H can be formed by allowing H to “roam free” in CRN -- it will fix the strains for you!
- Nonlocal changes accrue for changes in charge state for localized electrons/hole.
- Holes seem to stimulate more network response than electrons.

# Conclusions

- We need systematics: a set of reference models with “reference defects”, and the response of the system.
  - Need ways around the short time dynamics (Mousseau, ART; Parrinello scheme?).
  - A complement to the Chris paradigm learn what you can about a-Si from c-Si.